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Magnetic Spectroscopy of Functional Materials

Daniel Haskel

*Magnetic Materials Group, XOR
Advanced Photon Source, Argonne National Laboratory*

Outline

- Magnetic contrast with CP x-rays
- The needs and opportunities
- Best permanent magnets [PRL 95, 217207 (2005)]
- Giant magnetocaloric materials [PRL 98, 247205 (2007)]

Presented to *The University of Chicago Science Review Committee*
September 18, 2007.

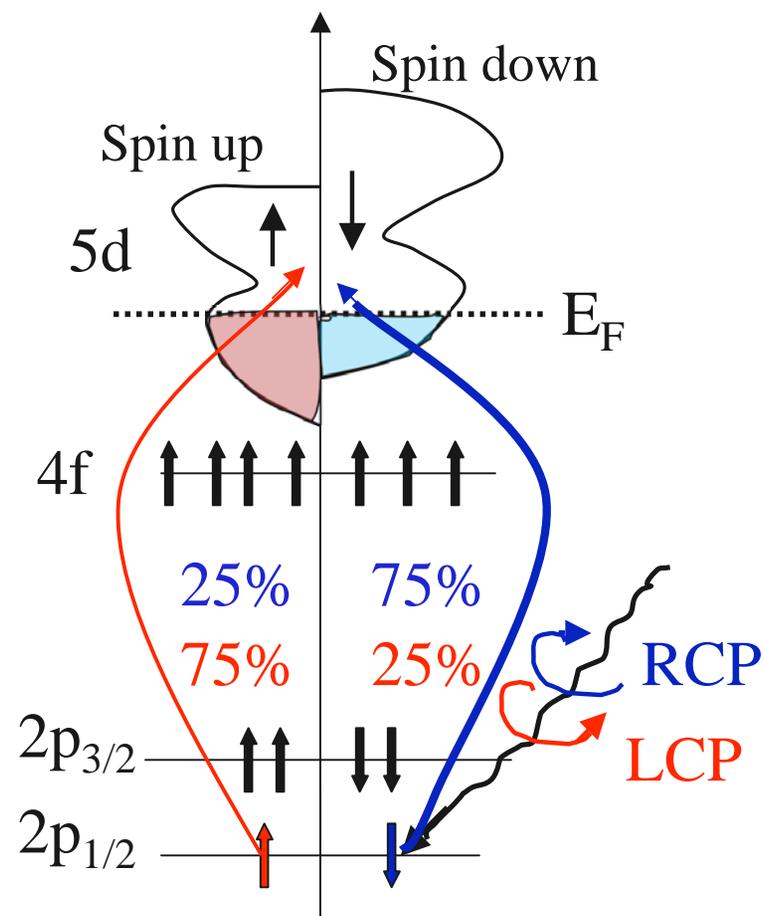
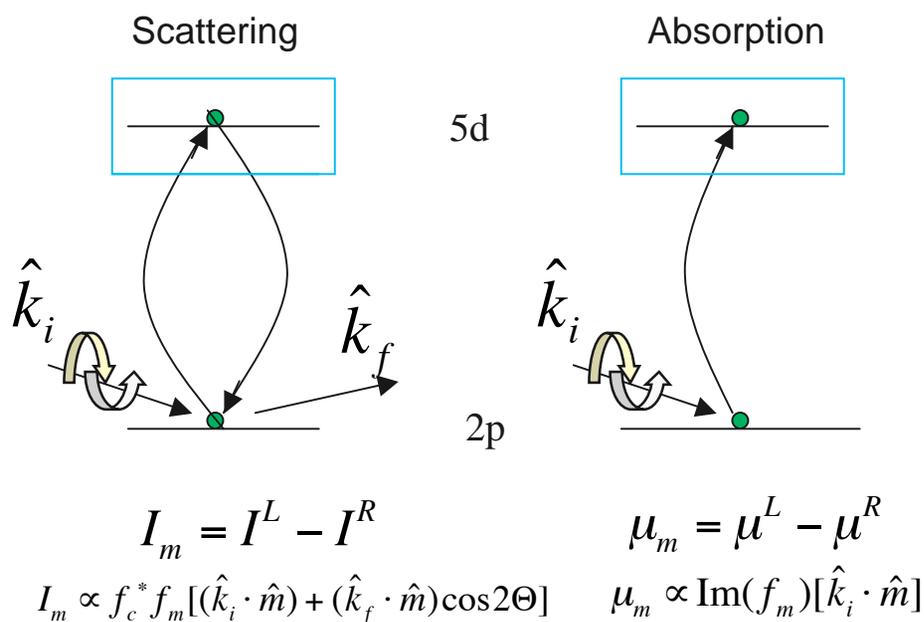
Magnetic Spectroscopy of Functional Materials

- → Magnetic contrast with CP x-rays
- The needs, the tools
- Best permanent magnets
- Giant magnetocaloric materials

Magnetic contrast with CP x-rays

Fano 1969; Schutz 1987

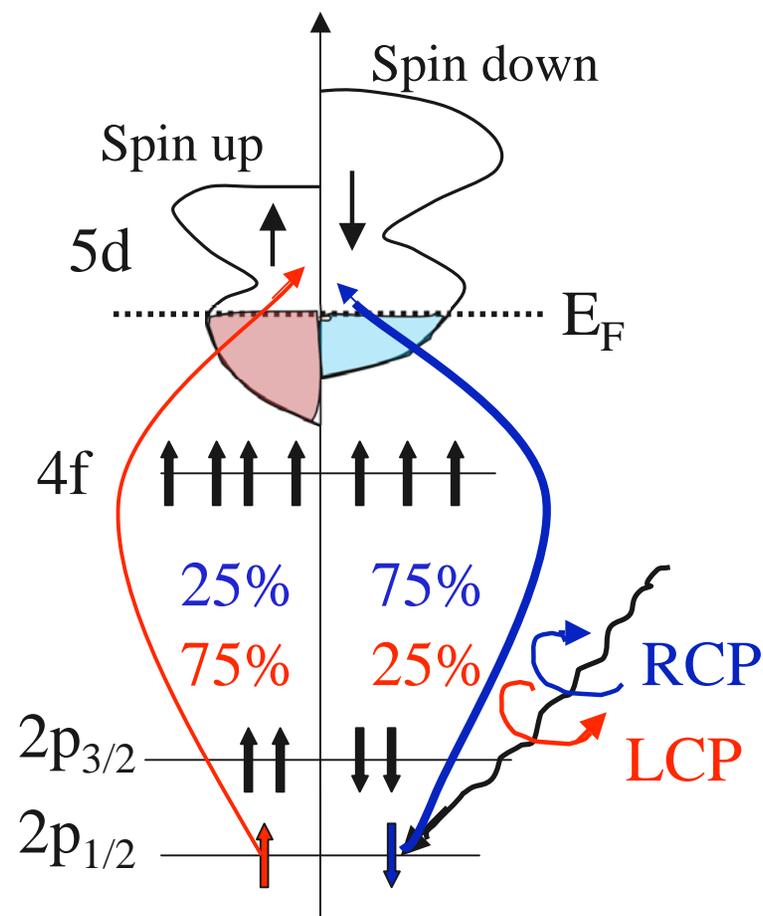
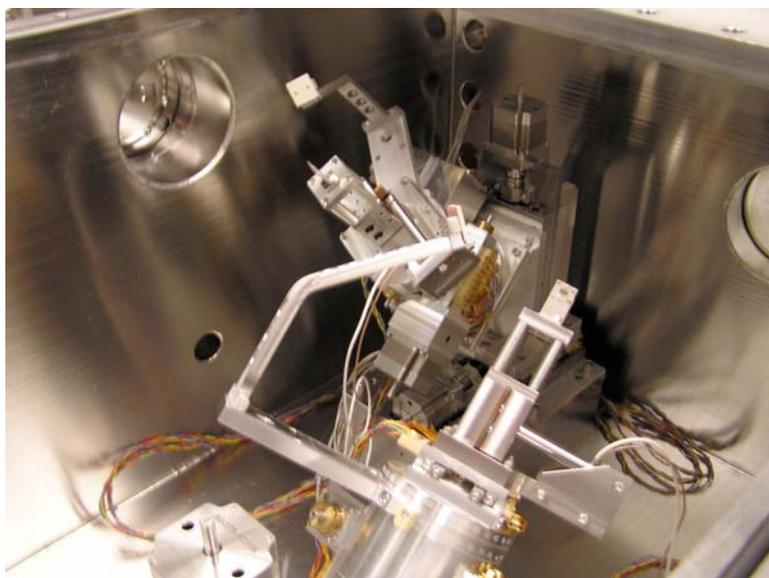
- Element- and orbital specific (resonance)
- Separation of S_z , L_z (sum rules)
- Resonant scattering adds Q dependence



Magnetic contrast with CP x-rays

Fano 1969; Schutz 1987

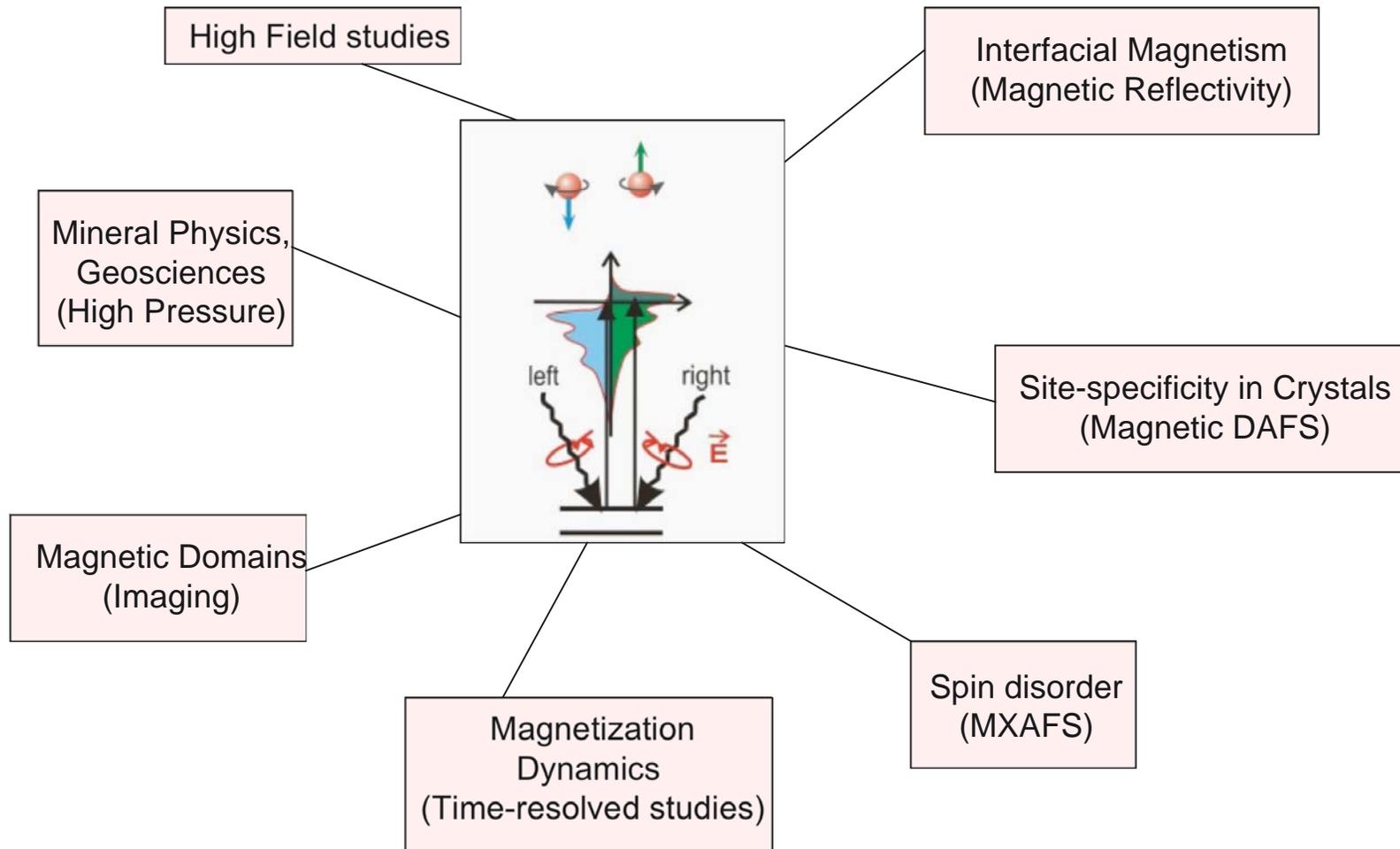
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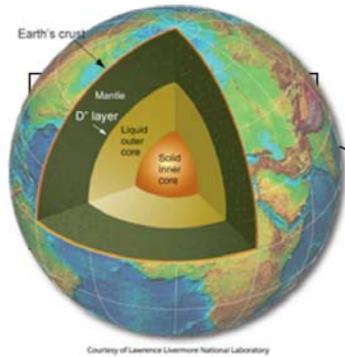


Magnetic Spectroscopy of Functional Materials

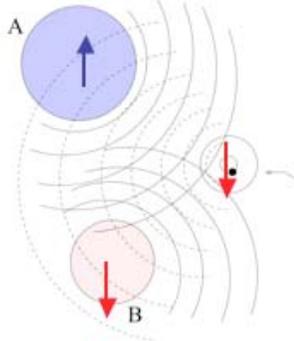
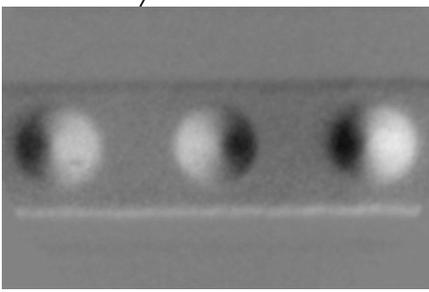
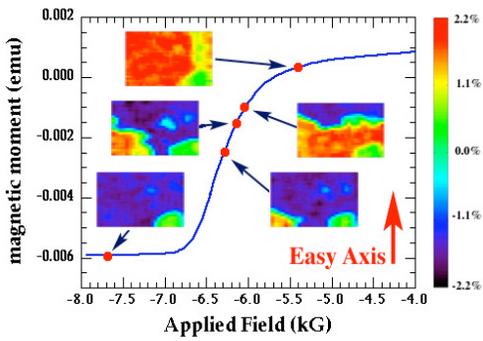
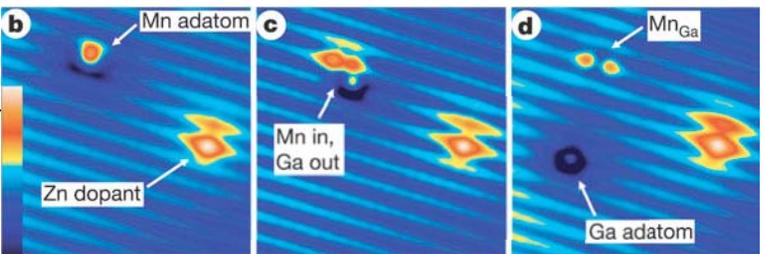
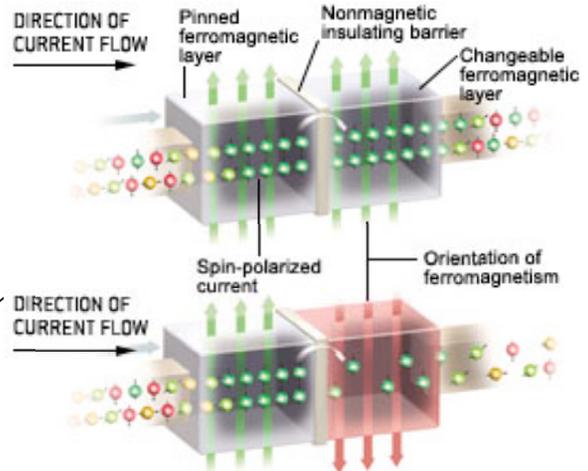
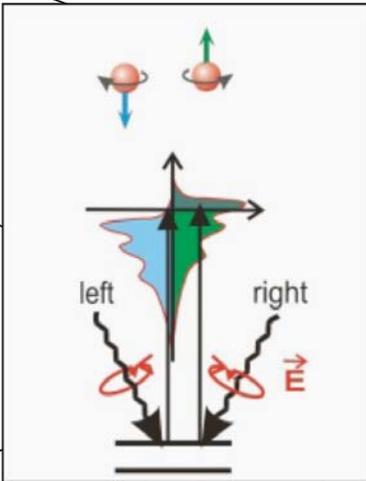
- Magnetic contrast with CP x-rays
- → The needs and opportunities
- Best permanent magnets
- Giant magnetocaloric materials

The needs and opportunities, the tools





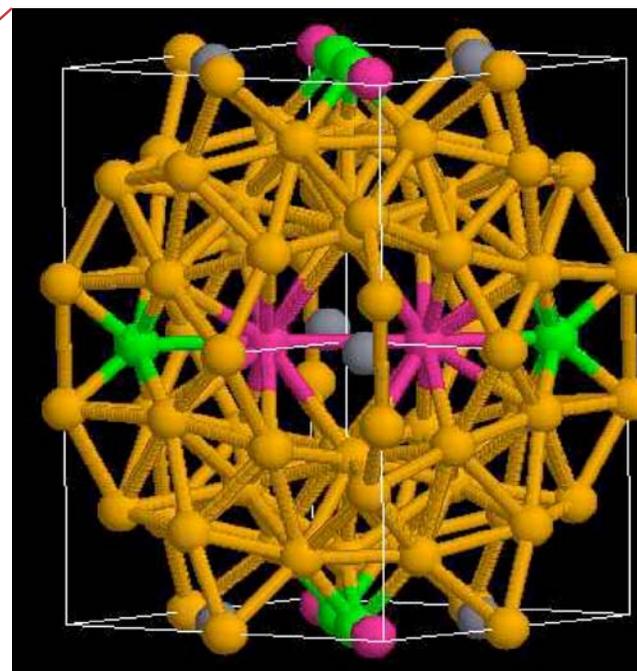
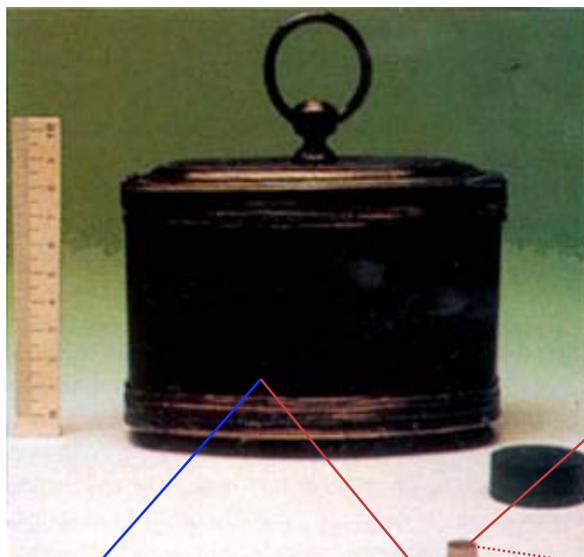
Courtesy of Lawrence Livermore National Laboratory



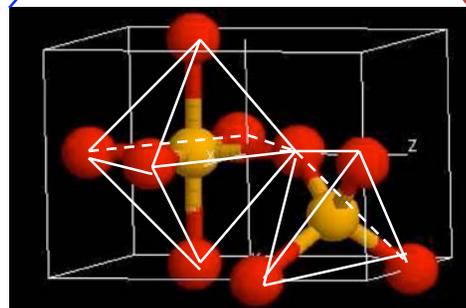
Magnetic Spectroscopy of Functional Materials

- Magnetic contrast with CP x-rays
- The needs and opportunities
- → **Best permanent magnets**
- Giant magnetocaloric materials

Permanent Magnets: Evolution brings complexity



$\text{Nd}_2\text{Fe}_{14}\text{B}$
(artificial)



Magnetite

● O
● Fe



● Fe ● Nd1 ● B
● Nd2

64 magnetic atoms
2 magnetic elements (Fe, Nd)

Modern magnets and the Rare-Earth role

Recipe for a good magnet:

- Large magnetization (pack high density of magnetic ions).
- Large coercivity (“magnetic hardness”, add rare-earth ions).

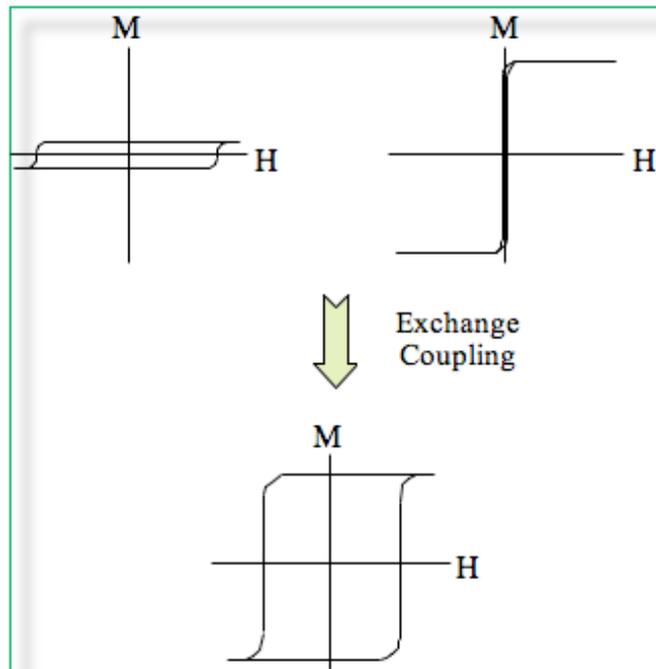
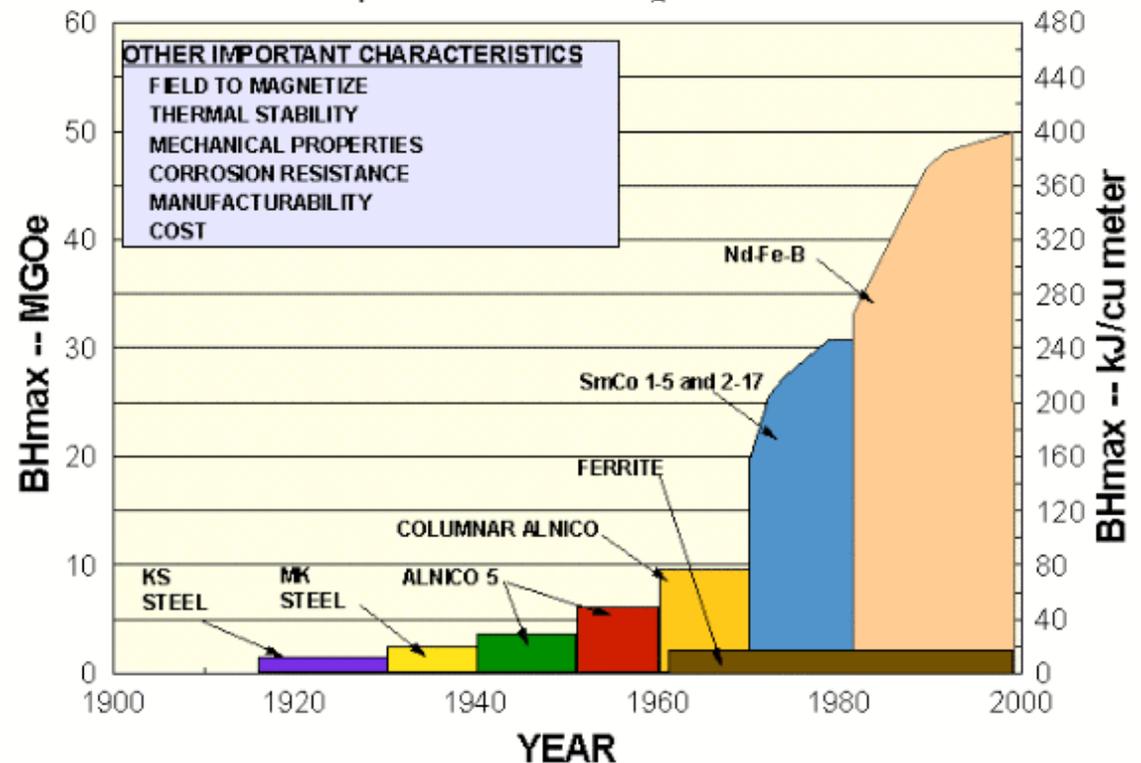


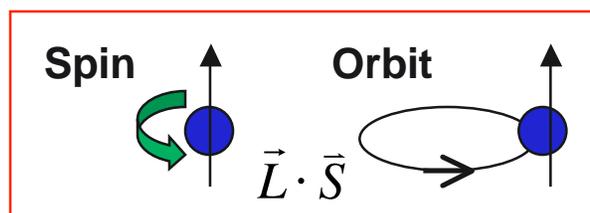
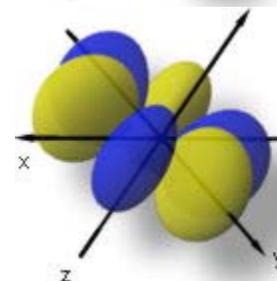
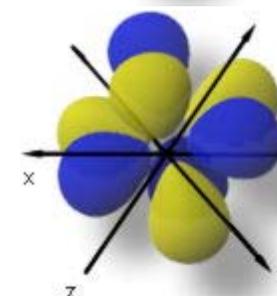
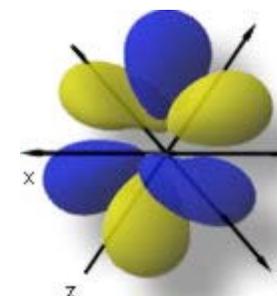
Figure I
Development of Permanent Magnets in the 1900's



Modern magnets and the Rare-Earth role

1	H																	2	He																
3	Li	4	Be											5	B	6	C	7	N	8	O	9	F	10	Ne										
11	Na	12	Mg											13	Al	14	Si	15	P	16	S	17	Cl	18	Ar										
19	K	20	Ca	21	Sc	22	Ti	23	V	24	Cr	25	Mn	26	Fe	27	Co	28	Ni	29	Cu	30	Zn	31	Ga	32	Ge	33	As	34	Se	35	Br	36	Kr
37	Rb	38	Sr	39	Y	40	Zr	41	Nb	42	Mo	43	Tc	44	Ru	45	Rh	46	Pd	47	Ag	48	Cd	49	In	50	Sn	51	Sb	52	Te	53	I	54	Xe
55	Cs	56	Ba	L	Hf	72	Ta	73	W	74	Re	75	Os	76	Ir	77	Pt	78	Au	79	Hg	80	Tl	81	Pb	82	Bi	83	Po	84	At	85	Rn		
67	Fr	68	Ra	A																															
	L	57	La	58	Ce	59	Pr	60	Nd	61	Pm	62	Sm	63	Eu	64	Gd	65	Tb	66	Dy	67	Ho	68	Er	69	Tm	70	Yb	71	Lu				
	A	89	Ac	90	Th	91	Pa	92	U	93	Np	94	Pu	95	Am	96	Cm	97	Bk	98	Cf	99	Es	100	Fm	101	Md	102	No	103	Lr				

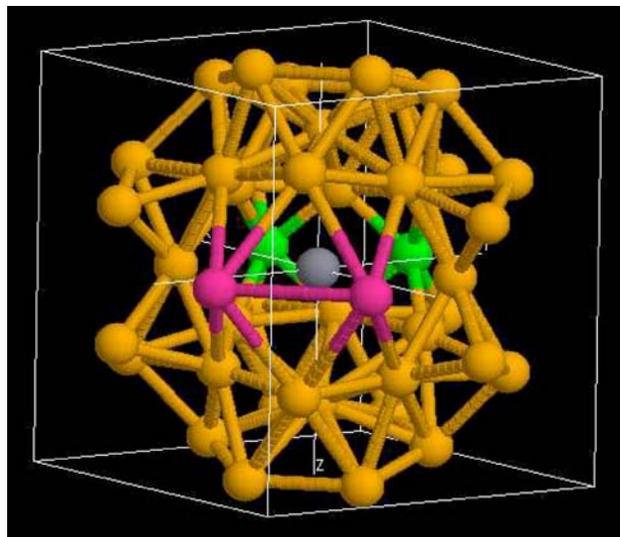
Rare-Earths:
4f electrons



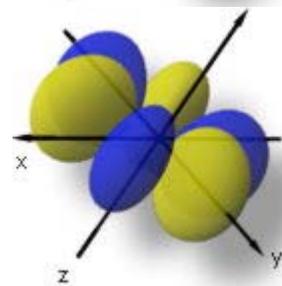
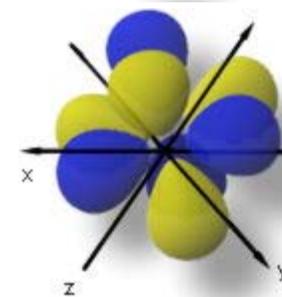
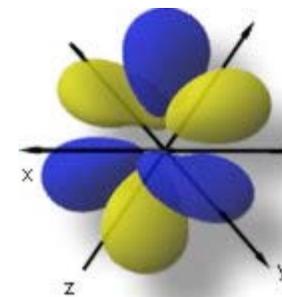
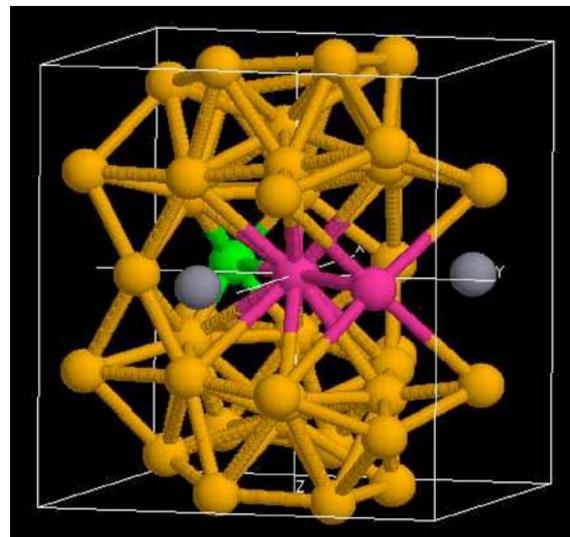
- *Aspherical* 4f orbitals interact with surrounding electrons (crystal field).
- *Spin-orbit coupling* + crystal field determine preferred spin orientation, “pinning” the magnetic moments.

$\text{Nd}_2\text{Fe}_{14}\text{B}$: *Best in its class*

- Fe: $\sim 31 \mu_{\text{B}}/\text{f.u.}$; Nd: $\sim 6 \mu_{\text{B}}/\text{f.u.}$ (element-specificity)
- Magnetocrystalline anisotropy [001] dominated by Nd RE ions.



- Nd g
- Nd f
- B
- Fe



Rare-earth Nd ions are simultaneously present in two different crystalline environments.

What are their roles?

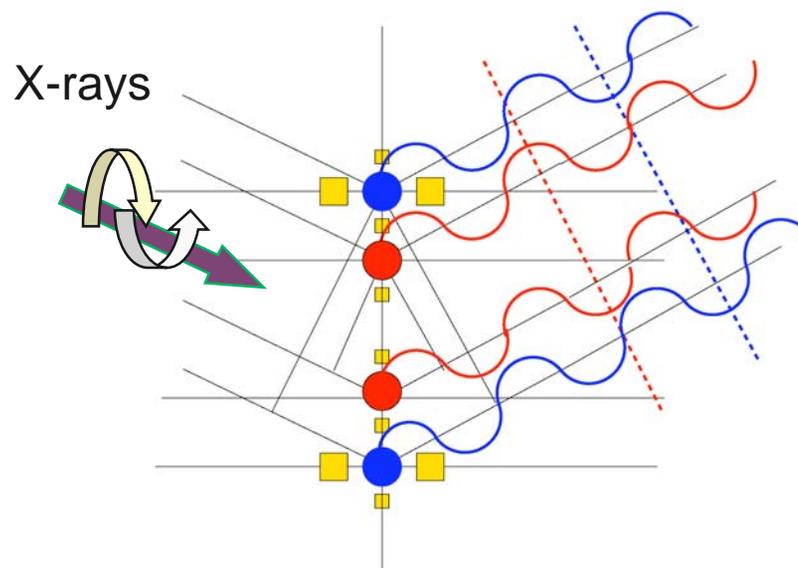
Exploiting the crystal's symmetry for site separation

$$F_{hkl} = f_{Nd}(Q, E) \sum_{i=1}^4 e^{i\vec{Q} \cdot \vec{r}_i}$$

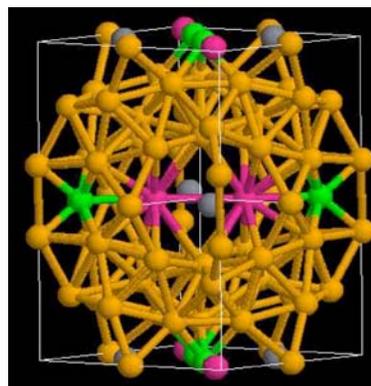
$$[\vec{Q} = 2\pi(\frac{h}{a}, \frac{k}{b}, \frac{l}{c}), \vec{r}_i = (x_i a + y_i b + z_i c)]$$

$$(4g) (x, \bar{x}, 0); (\bar{x}, x, 0); (\frac{1}{2} + x, \frac{1}{2} + x, \frac{1}{2}); (\frac{1}{2} - x, \frac{1}{2} - x, \frac{1}{2})$$

$$(4f) (x, x, 0); (\bar{x}, \bar{x}, 0); (\frac{1}{2} + x, \frac{1}{2} - x, \frac{1}{2}); (\frac{1}{2} - x, \frac{1}{2} + x, \frac{1}{2})$$

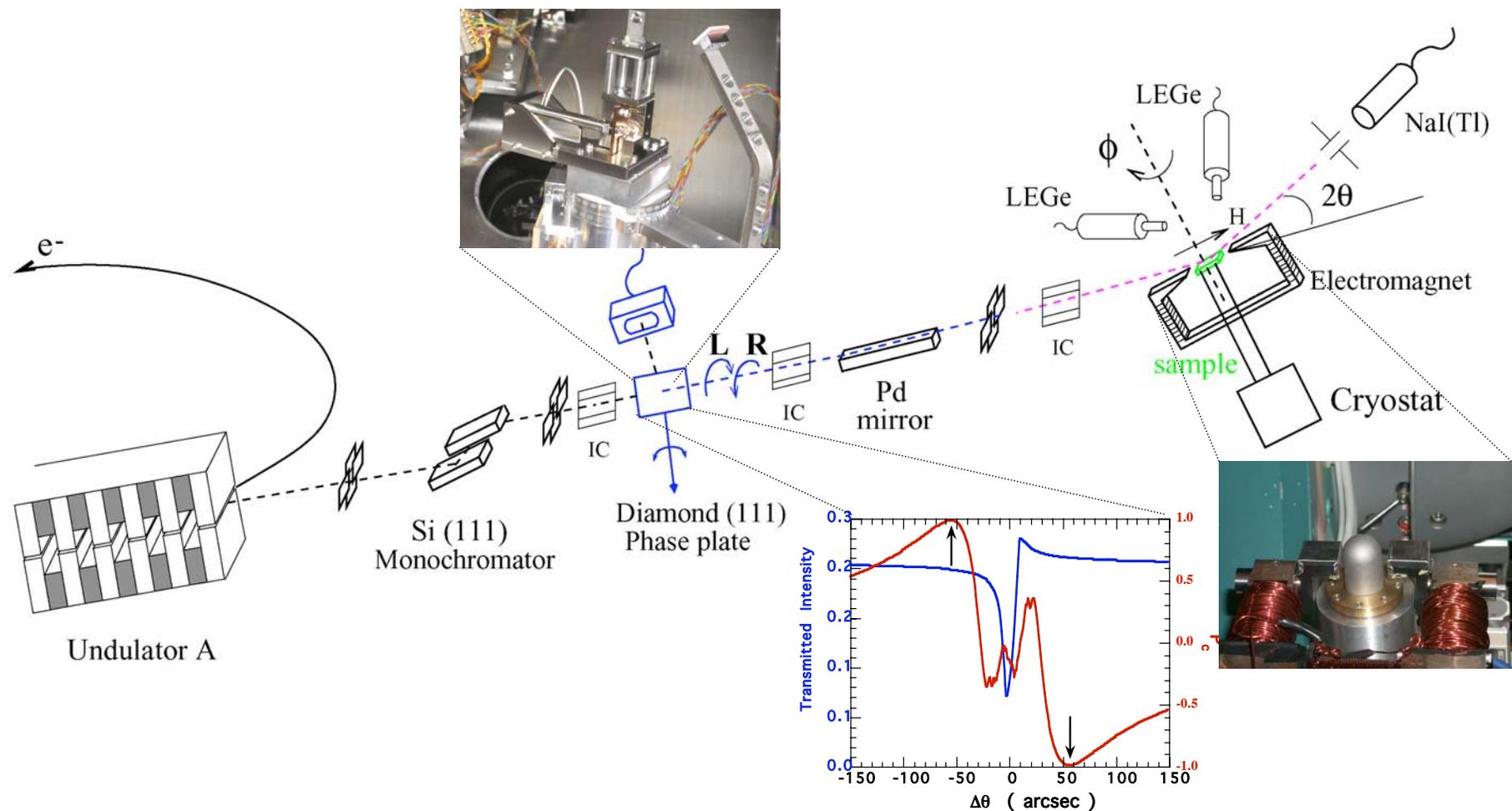


- Nd (4f)
- Nd (4g)
- B
- Fe



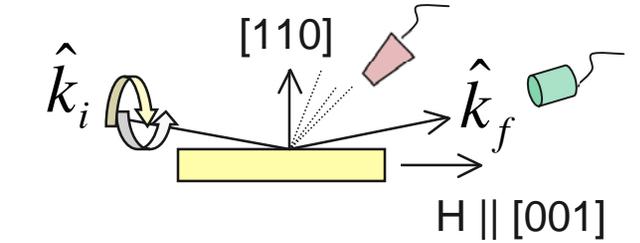
Site	(110)	(220)	(440)
Nd (4f)	3%	96.4%	48.5%
Nd (4g)	97%	3.6%	51.5%

Beamline 4-ID-D, Advanced Photon Source



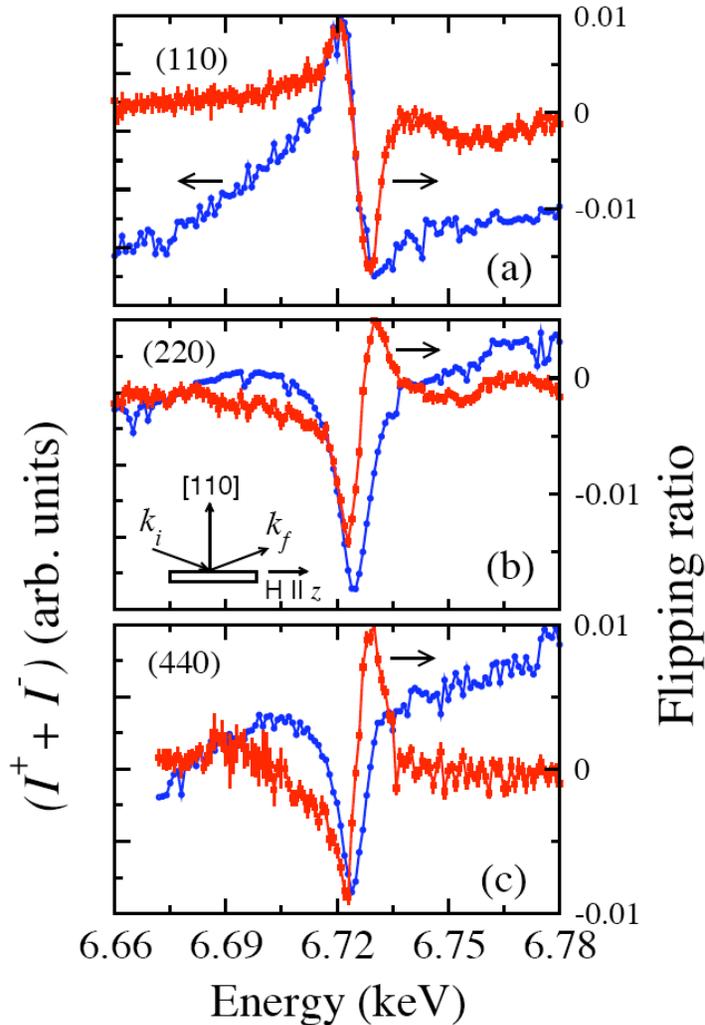
- Resonant diffraction at fix Q: scan ID gap, monochromator, phase plate, and sample angles with E. Digital lock-in detection of $(I^L - I^R)/(I^L + I^R)$.

Nd L_2 ($2p_{1/2}$)



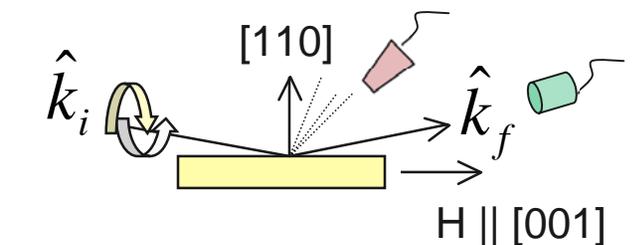
$$I_m \propto f_c^* f_m [(\hat{k}_i \cdot \hat{m}) + (\hat{k}_f \cdot \hat{m}) \cos 2\Theta]$$

Site	(110)	(220)	(440)
Nd (4f)	3%	96.4%	48.5%
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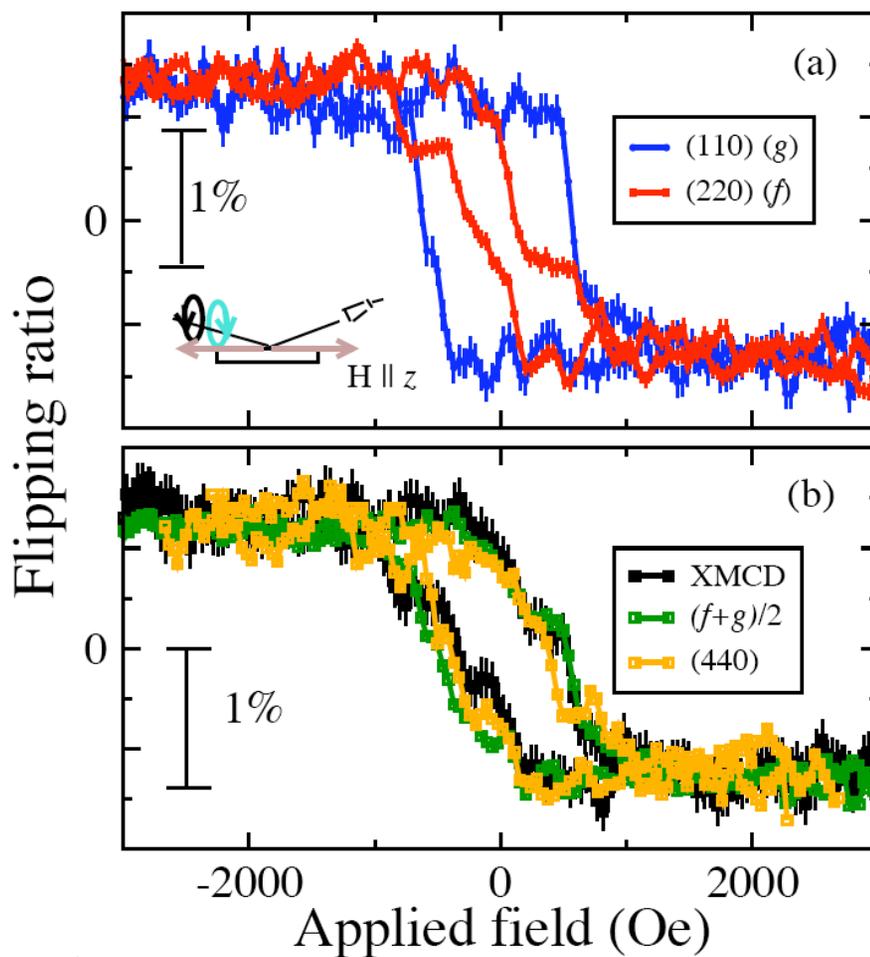
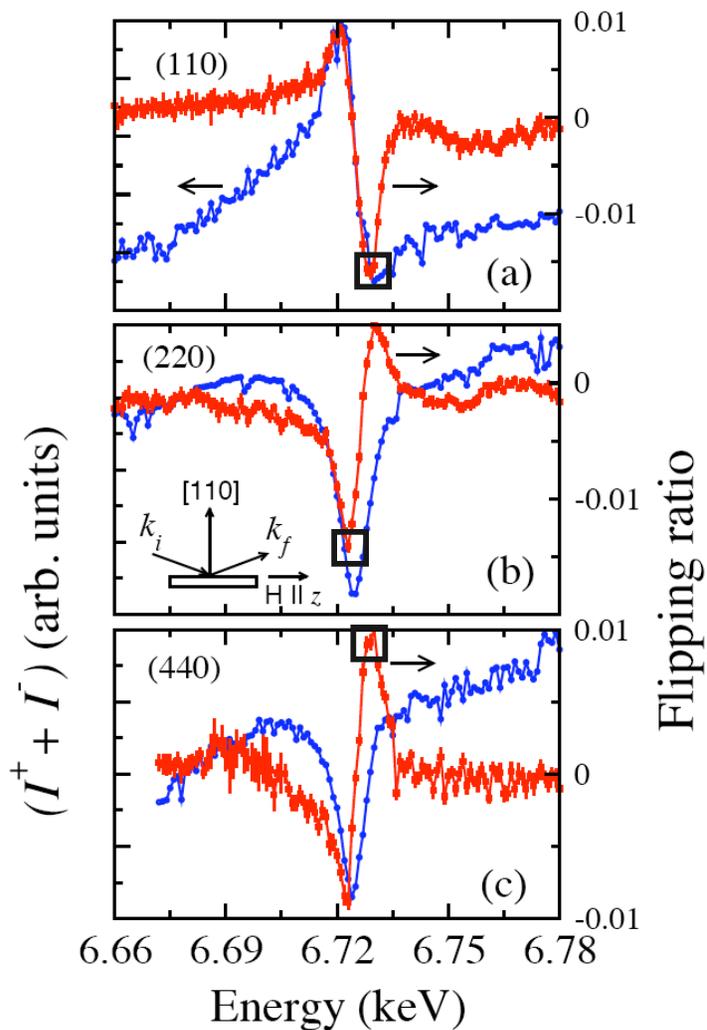
- (110) and (220) reflections probe element *and* site-specific magnetism.

Nd L_2 ($2p_{1/2}$)

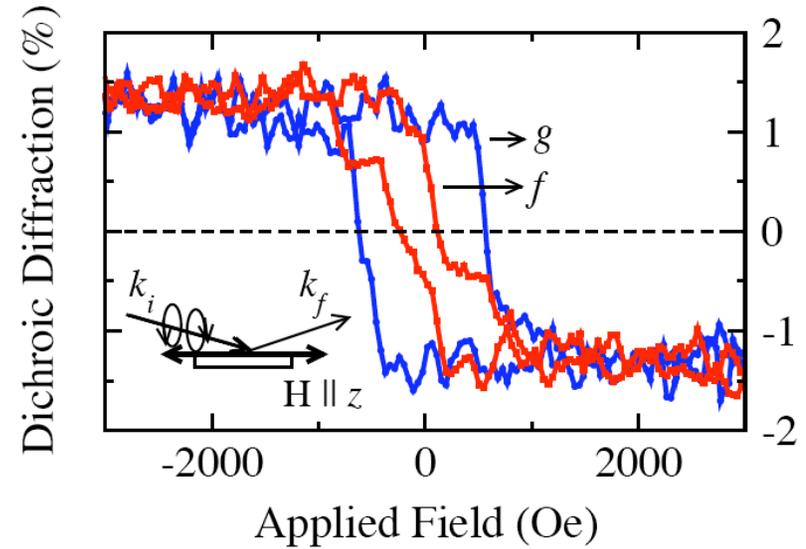
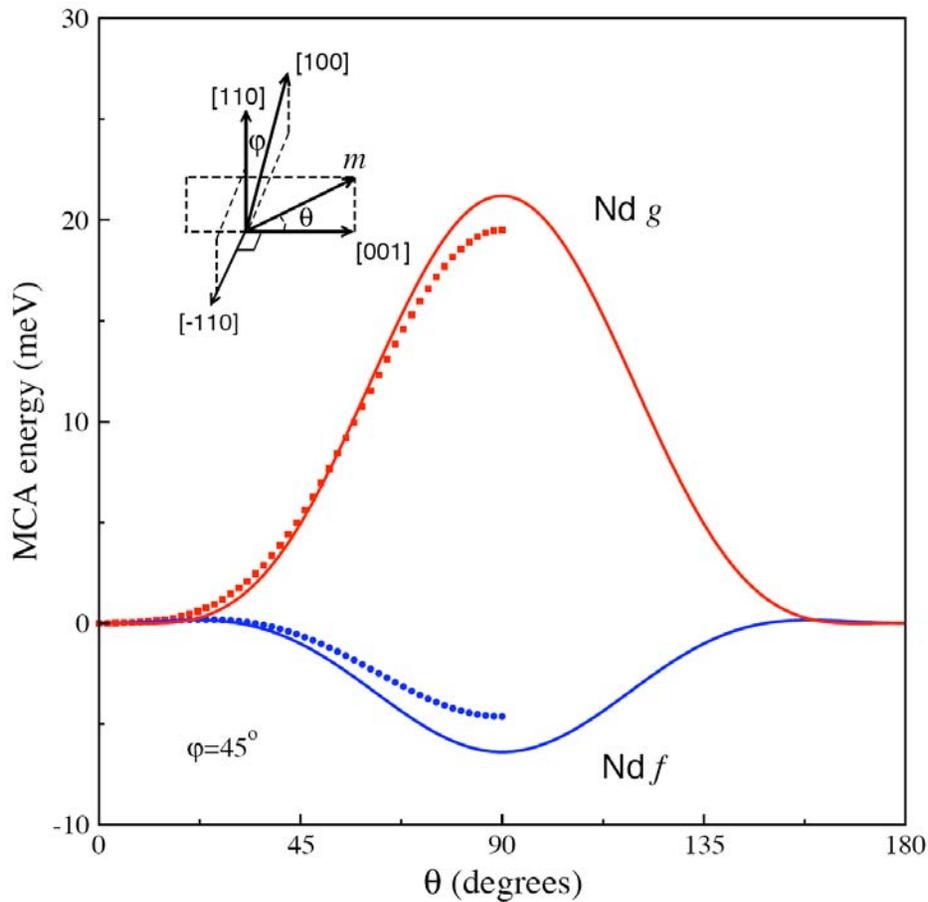


$$I_m \propto f_c^* f_m [(\hat{k}_i \cdot \hat{m}) + (\hat{k}_f \cdot \hat{m}) \cos 2\Theta]$$

Site	(110)	(220)	(440)
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Theoretical calculations (M. van Veenendaal, NIU-ANL)

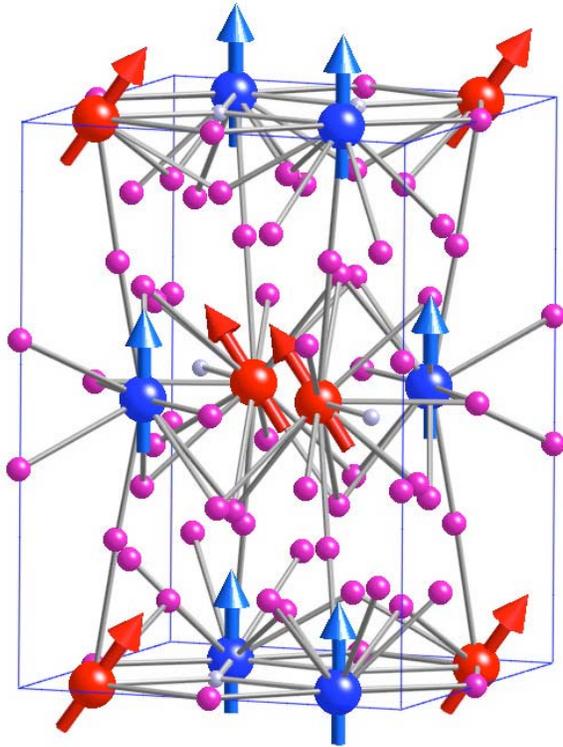


$$\langle LSJ, M = -J | H_{CEF} | LSJ, M = -J \rangle$$

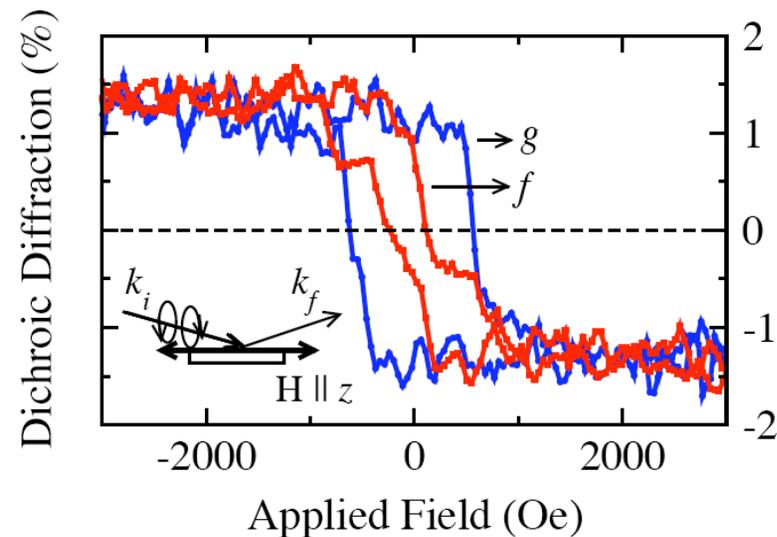
$$H_{CEF} = e \sum_i V(\vec{r}_i) = \sum_i B_{kq} (C_q^k)_i$$

CEF parameters M. Yamada *et al* (1988).

Manipulate magnetic hardness by atomic engineering



Phys. Rev. Lett. **95**, 217207 (2005)

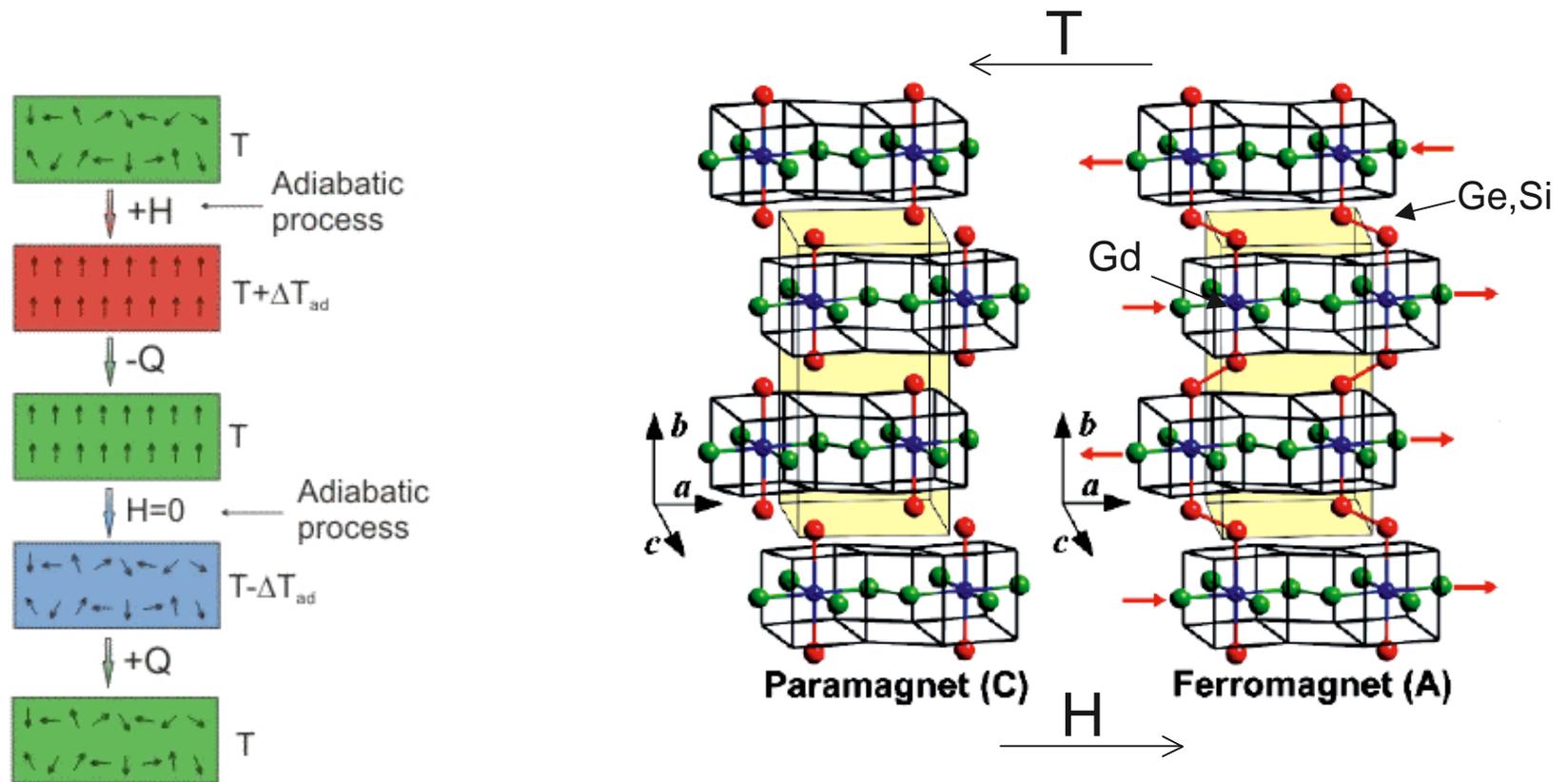


- Only one Nd site responsible for magnetic hardness.
- Replace “faulty” ions with other RE ions, or even Gd (isotropic).

Magnetic Spectroscopy of Functional Materials

- Magnetic contrast with CP x-rays
- The needs and opportunities
- Best permanent magnets
- → Giant magnetocaloric materials

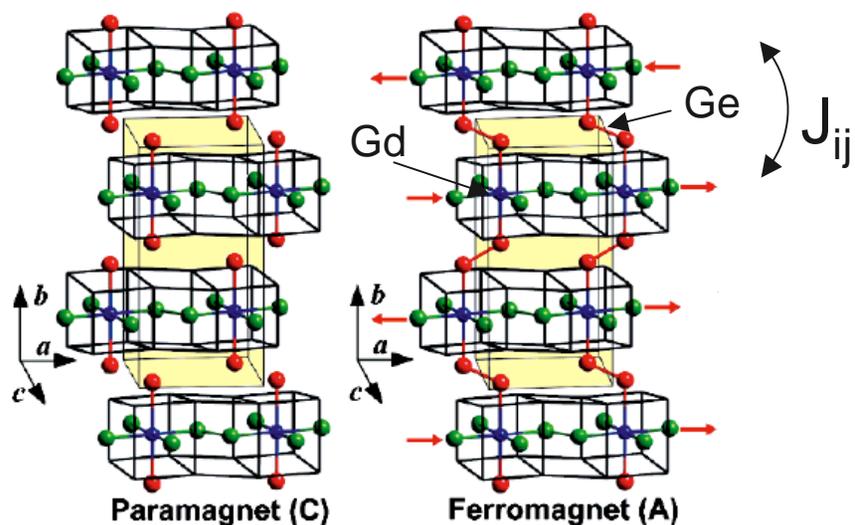
Giant magneto-caloric material $\text{Gd}_5(\text{Si}_x\text{Ge}_{1-x})_4$



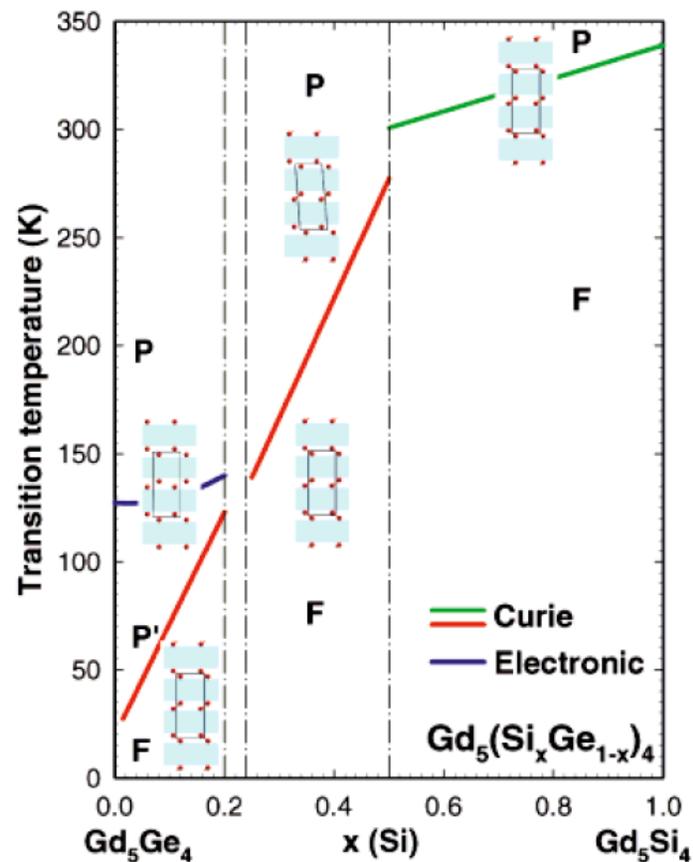
Pecharsky and Gschneider
Advanced Materials 13, 683 (2001).

- First-order magneto-structural transition yields large magnetic and structural entropy changes that can be harnessed for magnetic refrigeration.
- Transition temperature can be tuned with Si doping to near RT.

What is the role of Ge in mediating inter-layer exchange interactions?
 Why does Si doping enhance magnetic ordering?

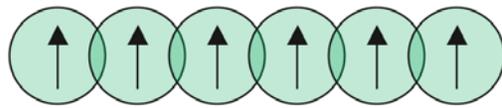


Pecharsky and Gschneider
Advanced Materials 13, 683 (2001).



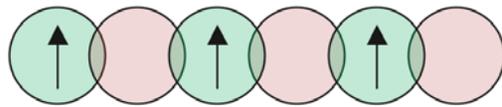
What is the role of Ge in mediating Gd-Gd exchange interactions?

- No direct exchange interaction between Gd 4*f* electrons (atomic-like).



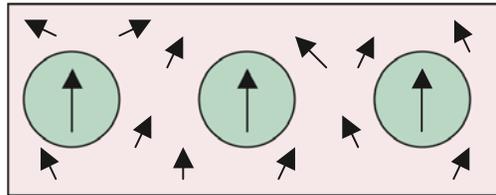
Direct Exchange
(overlap)

Pauli 1925, Heisenberg 1926



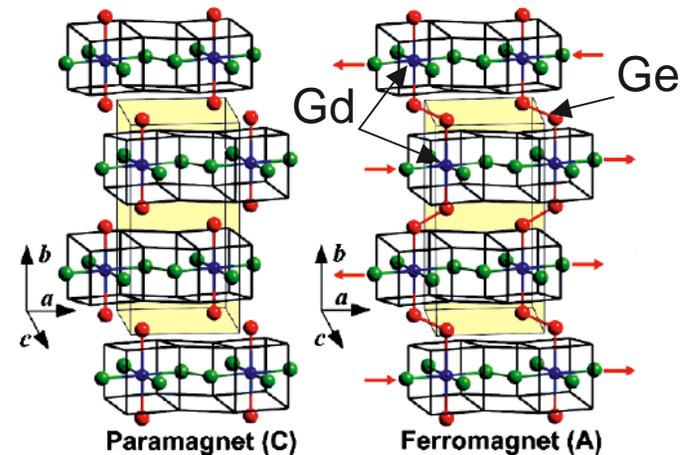
Superexchange
(overlap non-magnetic ion)

P.W. Anderson 1950



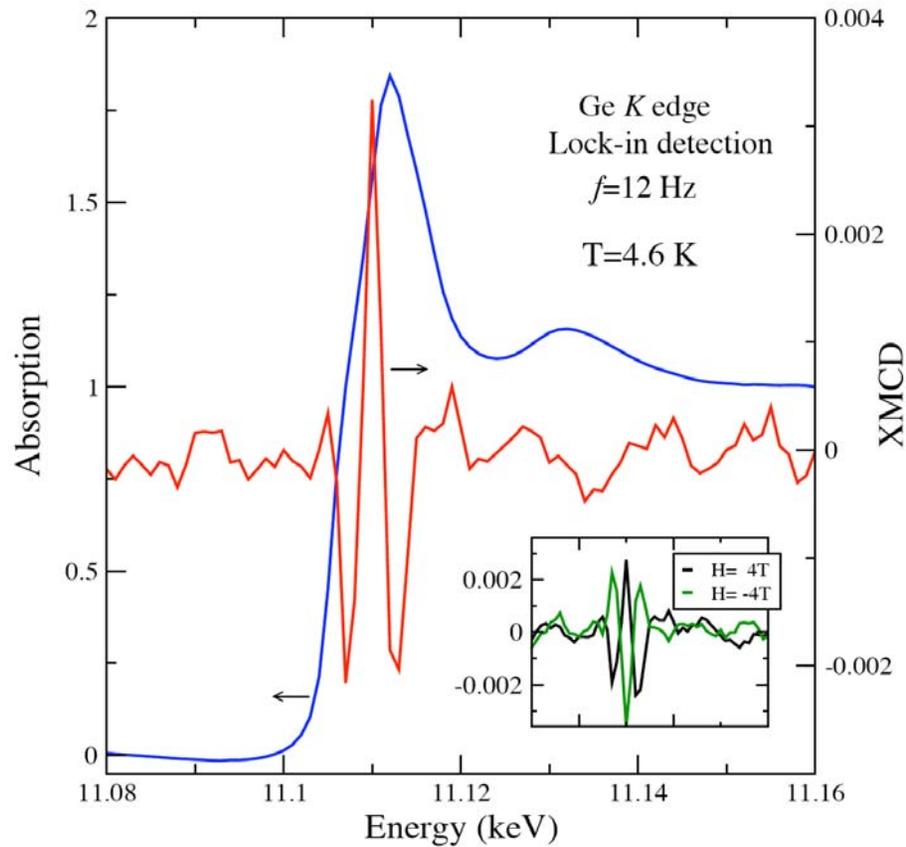
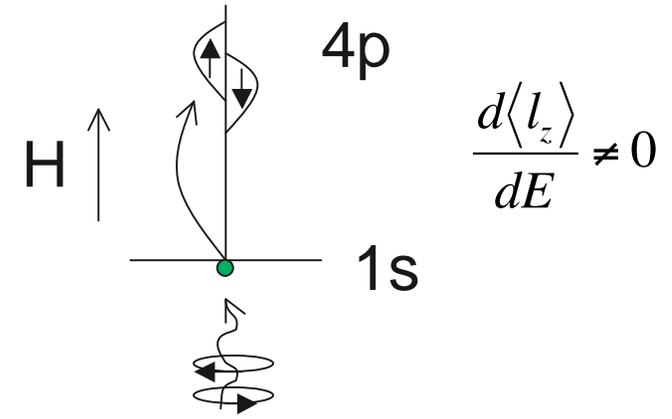
Indirect Exchange
(conduction electrons)

Ruderman and Kittel 1954



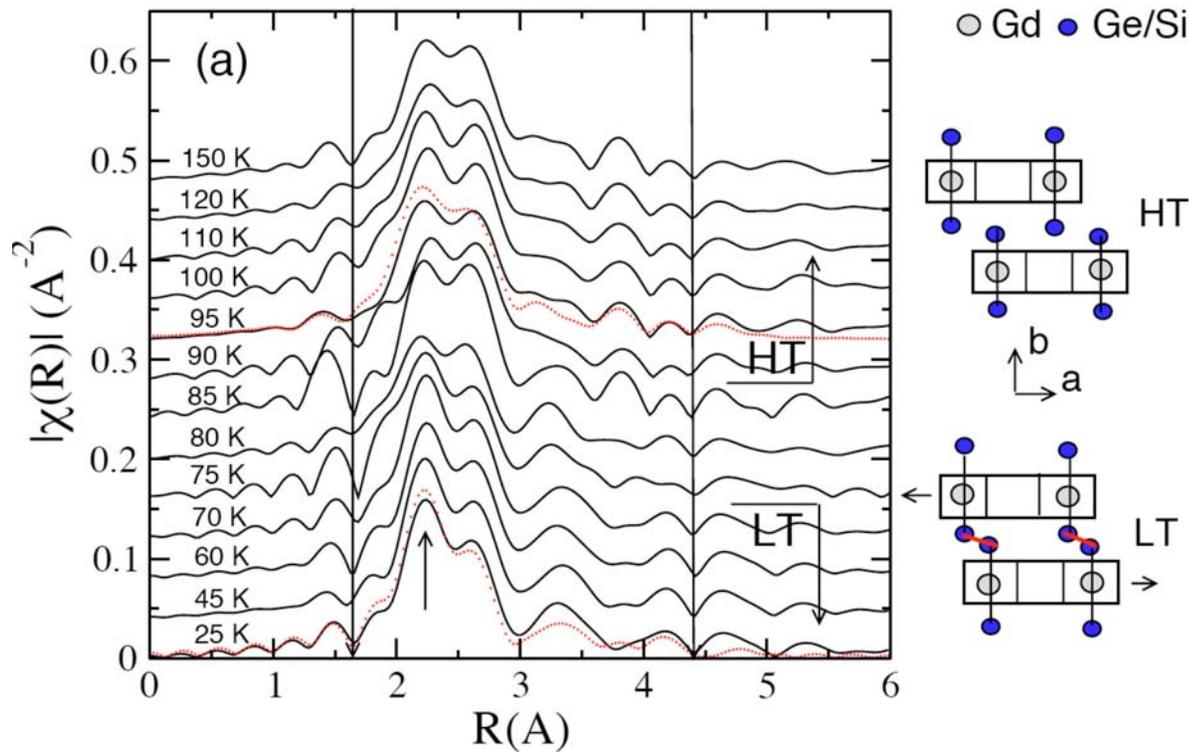
- Since Ge is non-magnetic, SE is believed to be responsible.

Is Ge *really* non-magnetic?

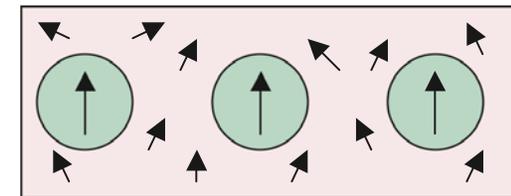
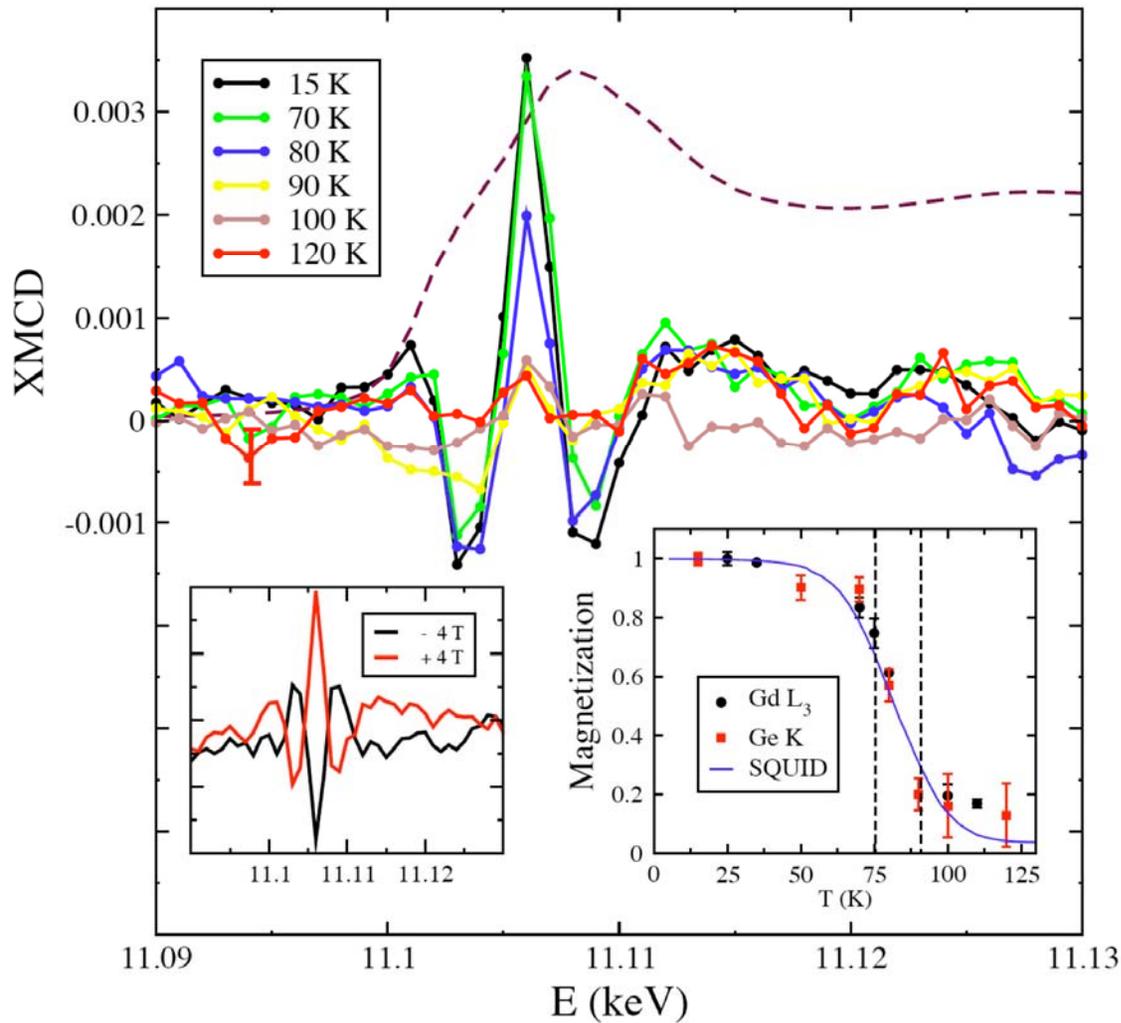


- Ge K- XMCD comparable to Fe,Co,Ni, but Ge is $3d^{10}$.

Is Ge magnetism tied to Gd magnetism and to the structural transition; i.e. intrinsic?

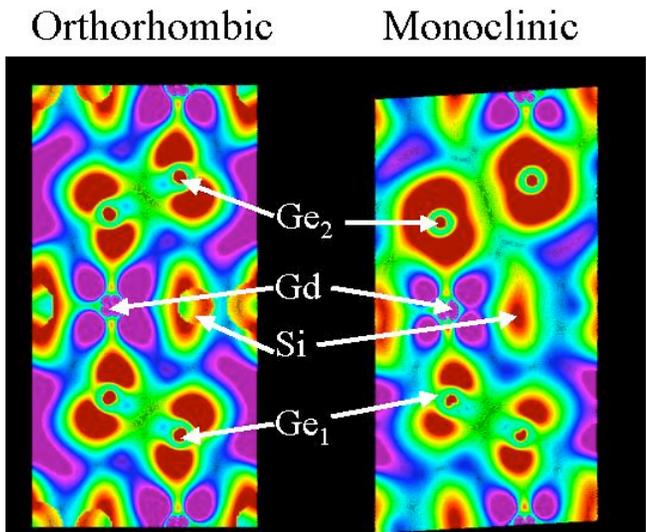
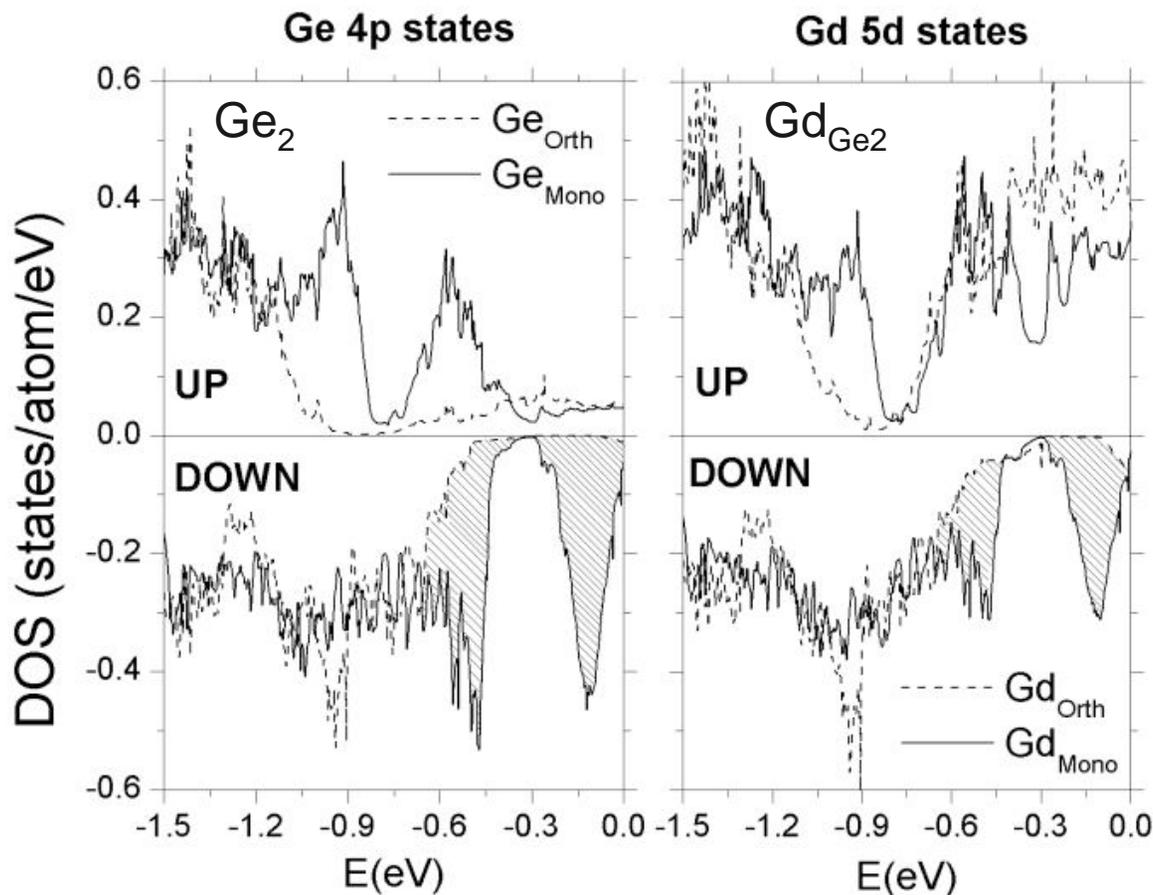


Is Ge magnetism tied to Gd magnetism and to the structural transition; i.e. intrinsic?



Gd 5d- Ge 4p hybridized band mediates indirect exchange?

DFT calculations (Y. B. Lee, B. Harmon- Ames)



S_z (μ_B) for Ge 4p, Gd 5d

	Ortho	Mono	Δ (%)
Ge_1	-0.033	-0.038	15%
Ge_2	-0.033	-0.06	100%
Gd_{Ge2}	+0.341	+0.23	30%

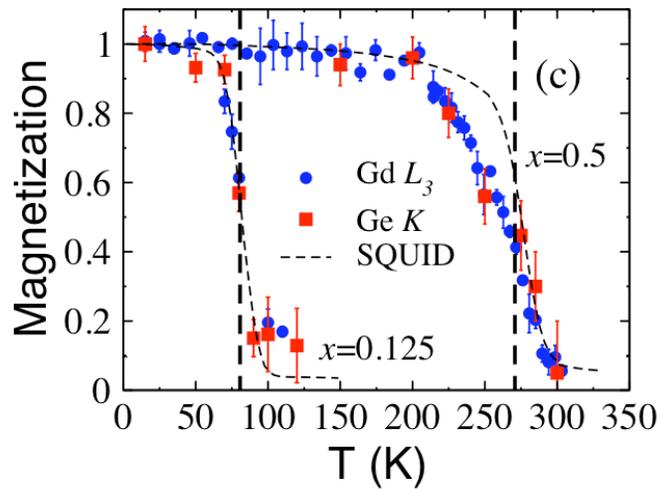
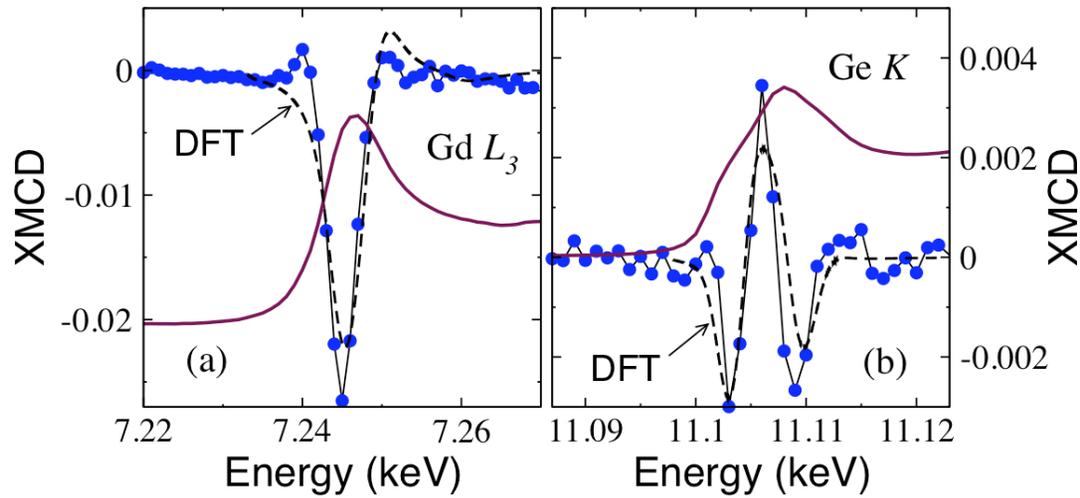
Bond breaking reduces the overlap of Gd 5d wavefunctions across disconnected slab

$$J_{ij} \propto \sum_k \langle P_{5d_i} \psi_k | P_{5d_j} \psi_k \rangle$$

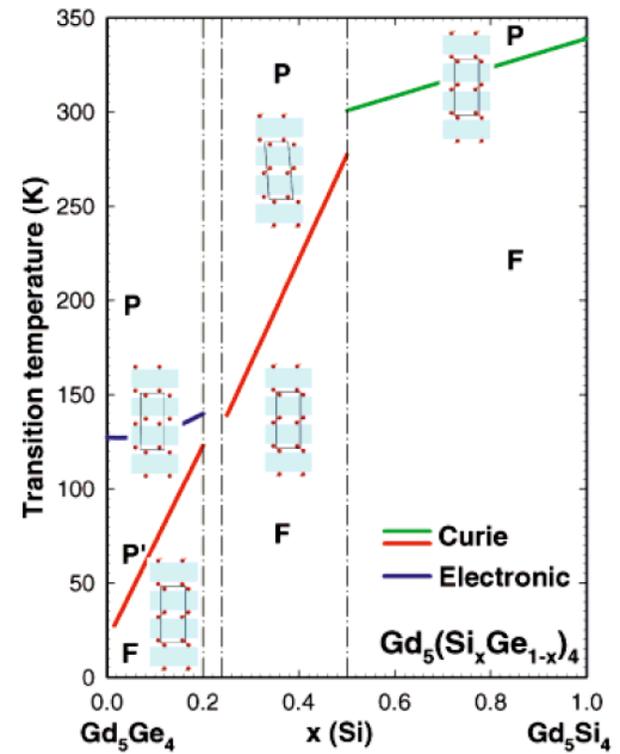
PRL 98, 247205 (2007)

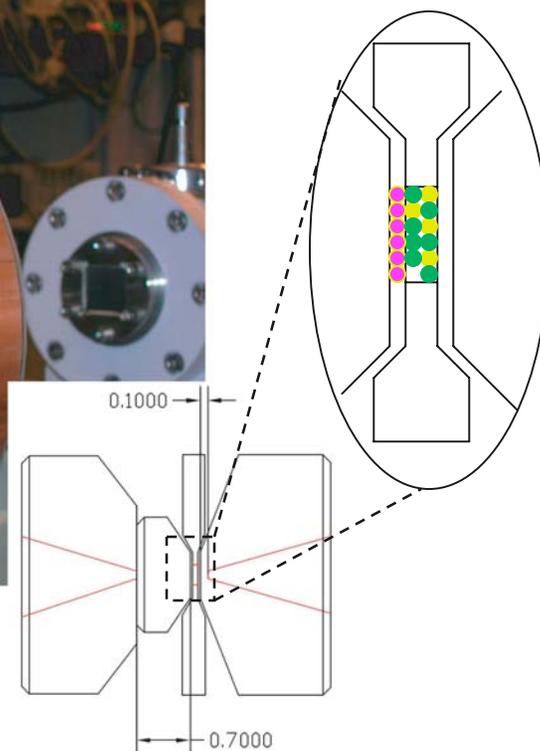
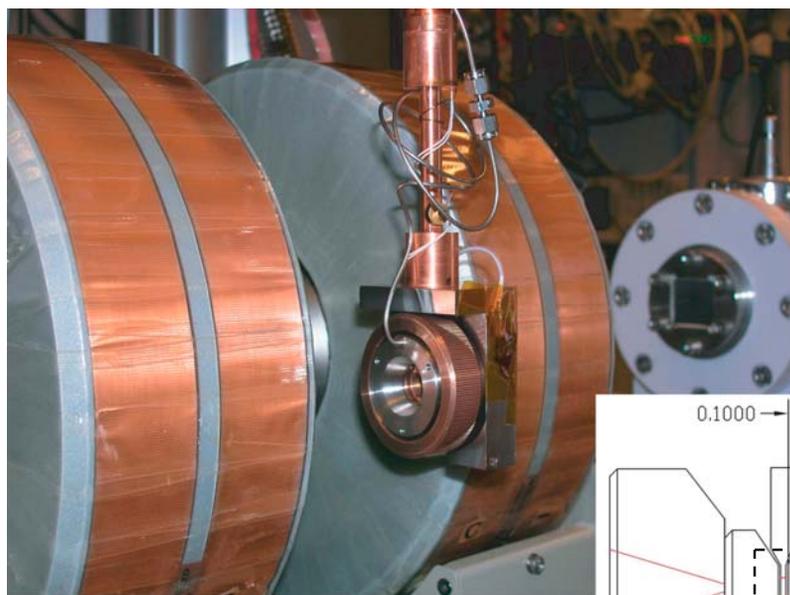
DFT calculations validated by XMCD data.

PRL 98, 247205 (2007)

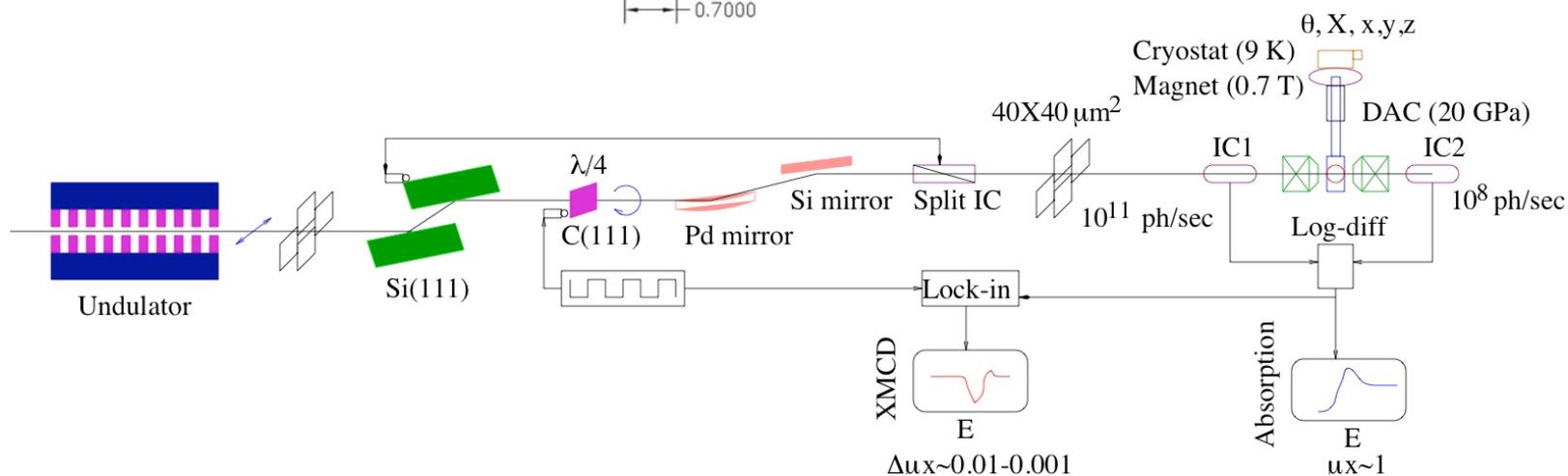


Why does Si doping enhance T_c ?

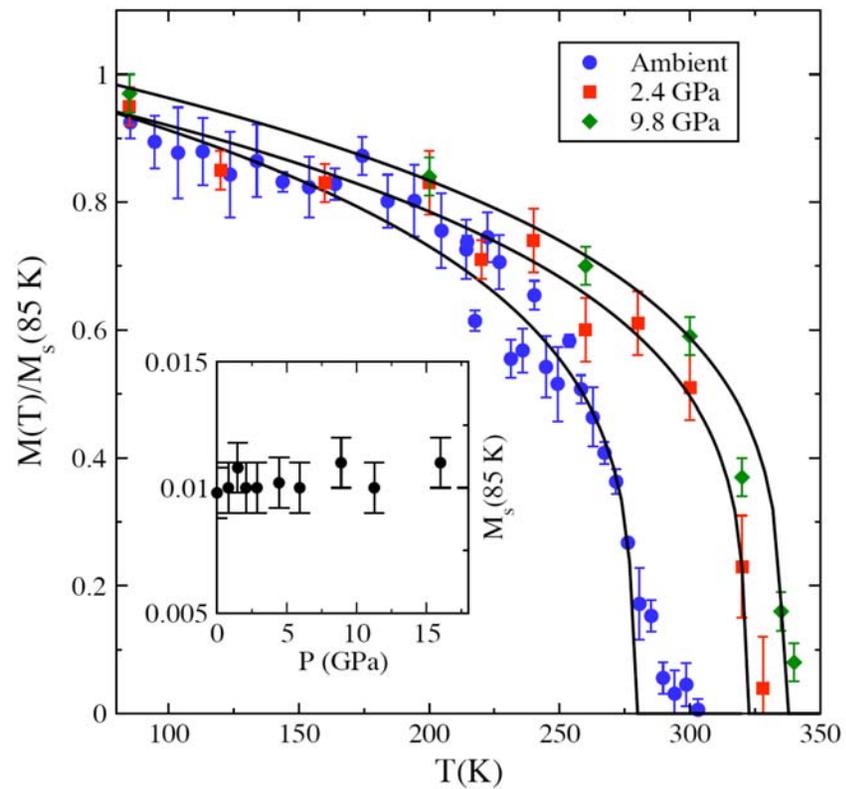
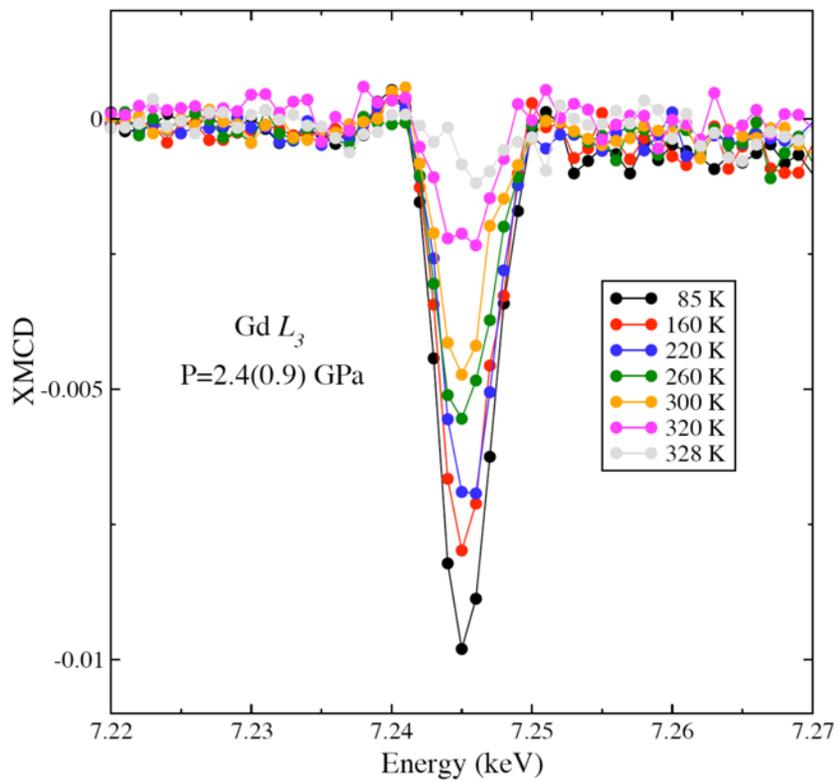


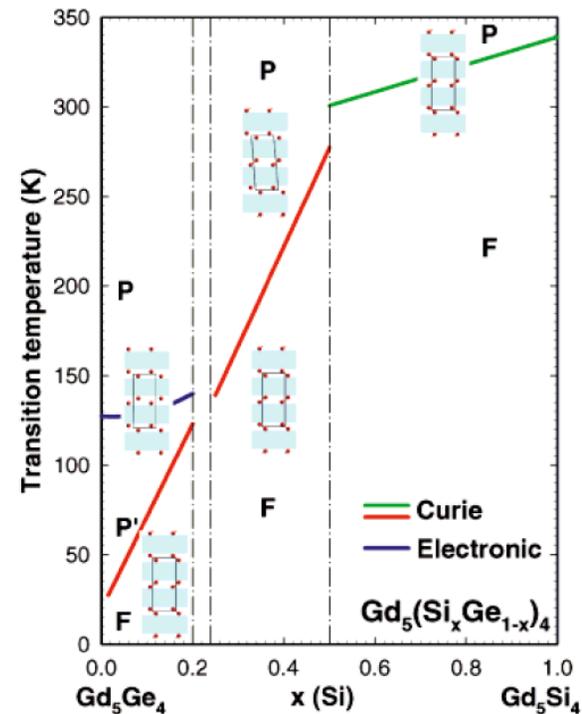
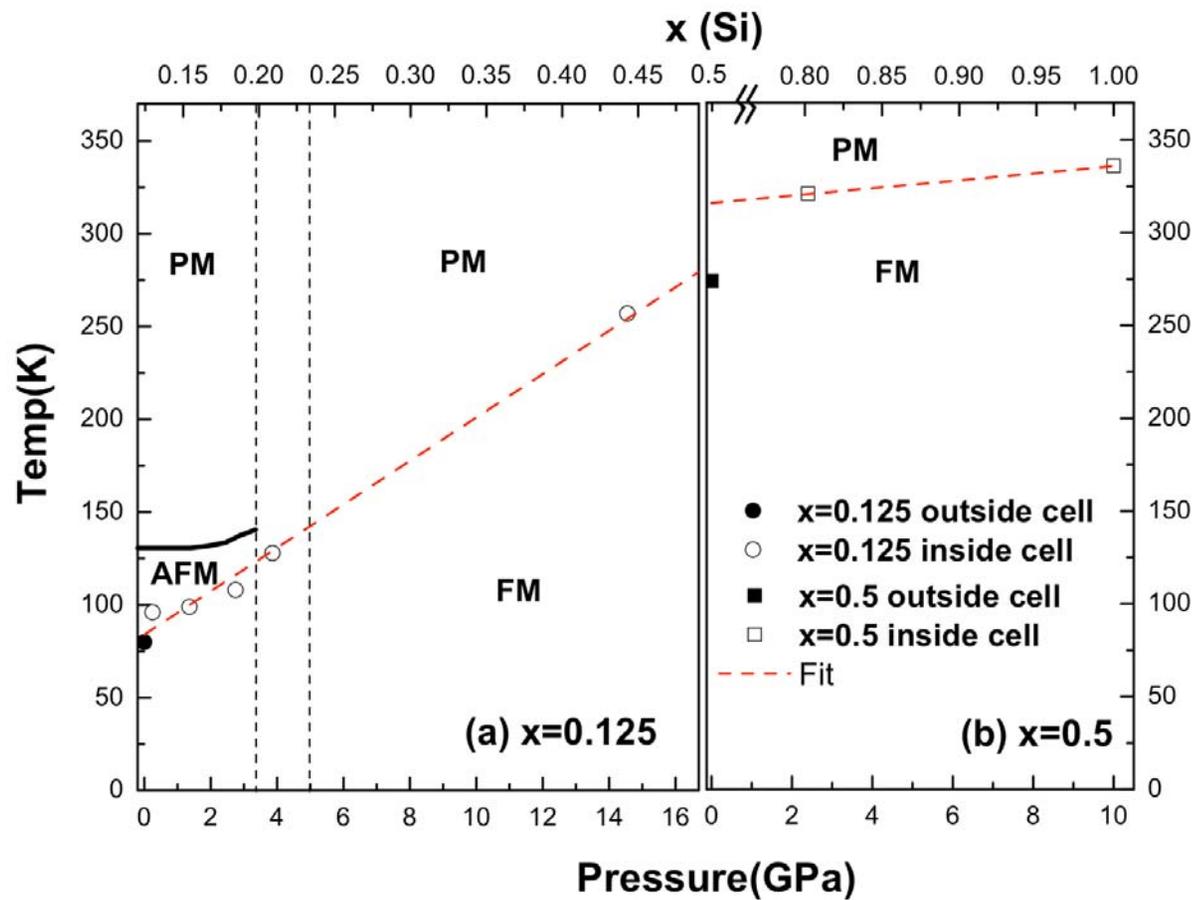


- GdSiGe
 - Cu
 - Ruby
- } Powder mixture
 In Si-oil (1:1:15 by volume)



Review of Scientific Instruments 78, 083094 (2007)



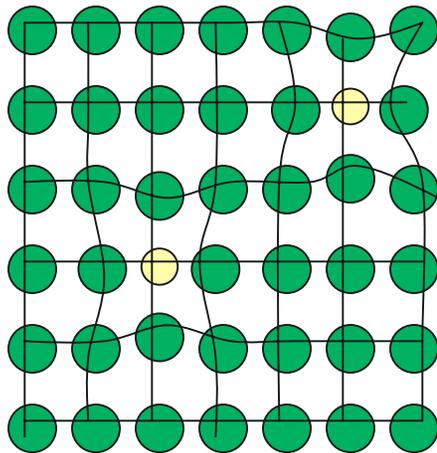


Enhanced magnetic order with Si doping is volume-driven
 Increased overlap enhances indirect exchange

Tseng *et al.*, PRB 76, 014411 (2007)

... but ... for the same macroscopic volume change T_c increases x3 faster with Si doping than with pressure...

$$(\Delta T_c / T_c)_{x,P} / (\Delta V / V)_{x,P}$$



● Gd/Ge
● Si

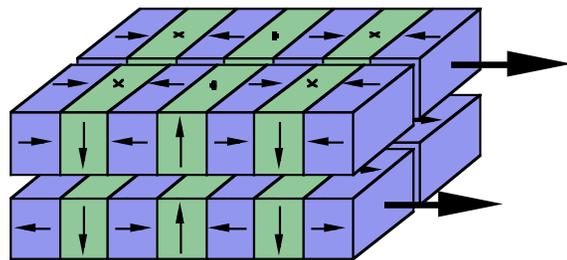
$$(\Delta V / V)_{local} > (\Delta V / V)_{macroscopic}$$

We postulate that compressed local regions around Si act as efficient FM exchange pathways and stabilize FM order faster than a smaller uniform compression of the lattice.

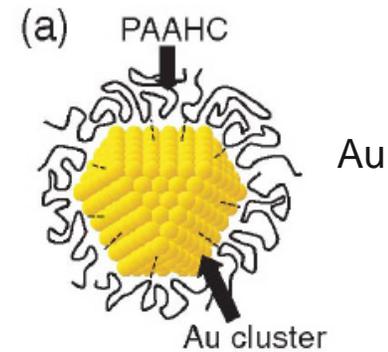
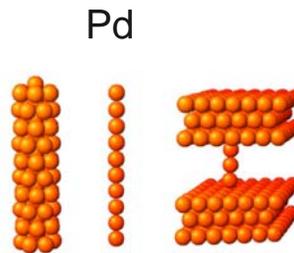
Tseng *et al.*, PRB 76, 014411 (2007)

Future directions

- Extend energy range of CP x-rays to 2.5-3.5 keV to include $L_{2,3}$ edges of 4d elements and $M_{2,3}$ edges of 5d elements.



Apple-type undulator



- Improve focusing capabilities to enable 1 Mbar XMCD experiments.

Collaborators

Y. Lee, B. Harmon, Y. Mudryk, V. Pecharsky, K. Gschneidner, P. Canfield
(Ames Laboratory, Iowa State University)

M. Van Veenendaal (Northern Illinois University and APS/ANL).

Y. C. Tseng, A. Cady, J. Lang, Z. Islam, G. Srajer, E. Kravtsov
(Magnetic Materials Group APS/ANL)

S. Sinogeikin (HPCAT- Carnegie Institute of Washington)